Effect of Morphological Idealization on Mean-Field Self-Consistent Homogenization of Polycrystals

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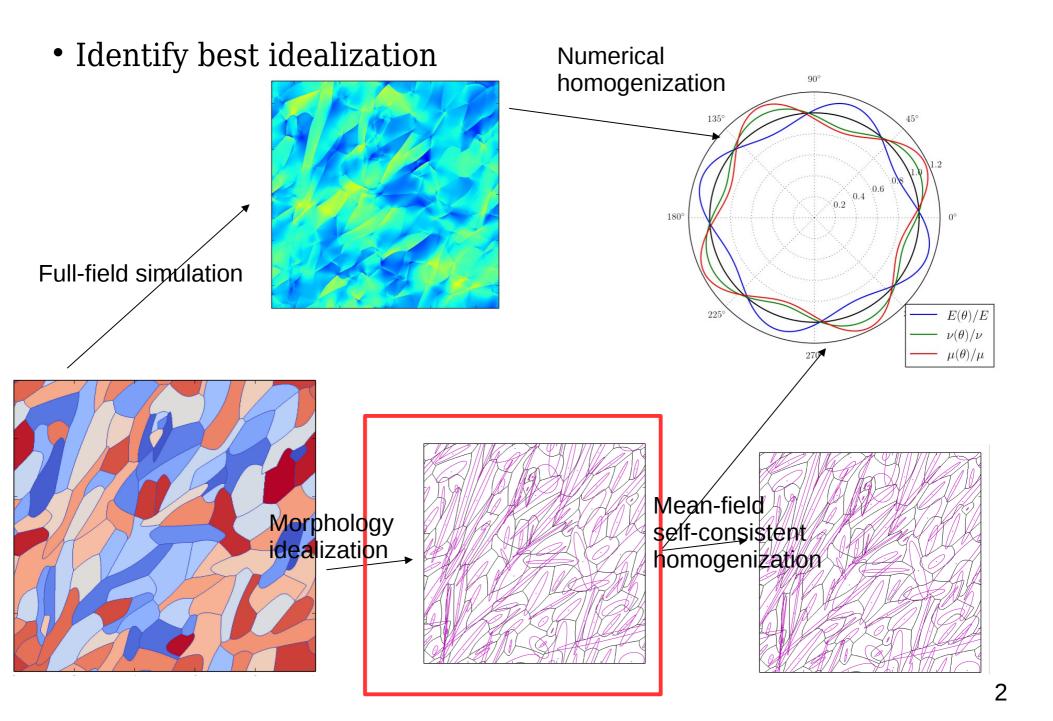
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Motivation and objective



Outline

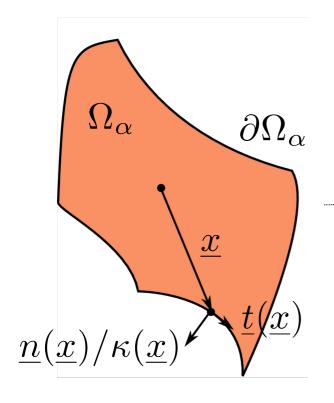
- Morphological characterization of single grains
- Numerical homogenization
- Self-consistent homogenization
- Shape idealization
- Comparison of SC estimates of elastic stiffness

Single grain morphology characterization

Single grains are characterized using Minkowski tensors:

Measures of mass distribution:

$$\mathcal{W}_0^{r,0} = \int_{\Omega_{\alpha}} \underline{x}^{\otimes^r} \mathrm{d}V$$

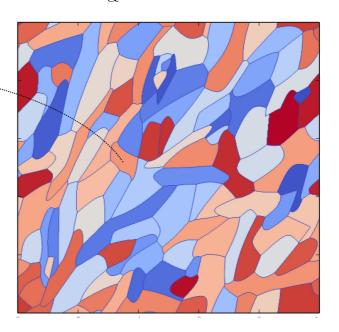


Measures of surface distribution:

$$\mathcal{W}_1^{r,s} = \int_{\partial\Omega_\alpha} \underline{x}^{\otimes^r} \odot [\underline{n}(\underline{x})]^{\otimes^s} dS$$

<u>Curvature-weighted measures of surface distribution:</u>

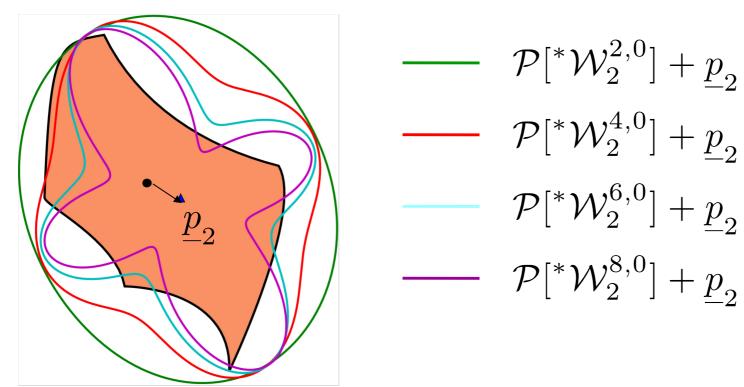
$$\mathcal{W}_{2}^{r,s} = \int_{\partial\Omega_{\alpha}} \kappa(\underline{x}) \underline{x}^{\otimes^{r}} \odot [\underline{n}(\underline{x})]^{\otimes^{s}} dS$$



Single grain morphology characterization

For a single grain, there are many different Minkowski tensors of a given order, most of which are independent.

Example: Reynolds glyphs of curvature-weighted tensors $\mathcal{W}_{2}^{r,0}$:



Question: What types and orders of Minkowski tensors are relevant and sufficient metrics for a grain?

Classic FFT solver

Following Moulinec and Suquet (1998), we have:

$$\begin{array}{llll} \mathbf{1} \ \widehat{\mathbf{N}}(\underline{k}) = [\underline{k} \, \mathbb{C}_0 \, \underline{k}]^{-1} & \forall \, \underline{k} \neq \underline{0} \\ \mathbf{2} \ \widehat{\mathbf{\Gamma}}(\underline{k}) = \underline{k} \odot \widehat{\mathbf{N}}(\underline{k}) \odot \underline{k} & \forall \, \underline{k} \neq 0, \ \widehat{\mathbf{\Gamma}}(\underline{0}) = \mathbb{O} \\ \mathbf{3} \ ^0 \boldsymbol{\varepsilon}(\underline{y}) = \langle \boldsymbol{\varepsilon} \rangle, \ ^0 \boldsymbol{\sigma}(\underline{y}) = \mathbb{C}(\underline{y}) : ^0 \boldsymbol{\varepsilon}(\underline{y}) \\ \mathbf{4} \ m = 0, \ \epsilon = 2\epsilon_{tol} \\ \mathbf{5} \ \text{while} \ \epsilon > \epsilon_{tol} : \\ \mathbf{6} \ & ^m \boldsymbol{\tau}(\underline{y}) = ^m \boldsymbol{\sigma}(\underline{y}) - \mathbb{C}_0 : ^m \boldsymbol{\varepsilon}(\underline{y}) \\ \mathbf{7} \ & ^m \Delta \widehat{\boldsymbol{\varepsilon}}(\underline{k}) = -\widehat{\mathbf{\Gamma}}(\underline{k}) : \mathcal{F}\{^m \boldsymbol{\tau}(\underline{y})\}(\underline{k}) \\ \mathbf{8} \ & ^{m+1} \boldsymbol{\varepsilon}(\underline{y}) = \langle \boldsymbol{\varepsilon} \rangle + \mathcal{F}^{-1}\{^m \Delta \widehat{\boldsymbol{\varepsilon}}(\underline{k})\}(\underline{y}) \\ \mathbf{9} \ & ^{m+1} \boldsymbol{\sigma}(\underline{y}) = \mathbb{C}(\underline{y}) : ^{m+1} \boldsymbol{\varepsilon}(\underline{y}) \\ \mathbf{10} \ & \epsilon = ^{m+1} \epsilon \end{array} \qquad \qquad \text{Version 1}$$

Discrete system:

$$x_{1} = \{iL_{x}/N_{x} \mid i = 0, \dots, N_{x} - 1\}$$

$$x_{2} = \{iL_{y}/N_{y} \mid i = 0, \dots, N_{y} - 1\}$$

$$k_{1} = \{\frac{i - N_{x}/2}{L_{x}} \mid i = 0, \dots, N_{x} - 1\}$$

$$k_{2} = \{\frac{i}{L_{x}} \mid i = 0, \dots, N_{y}/2\}$$

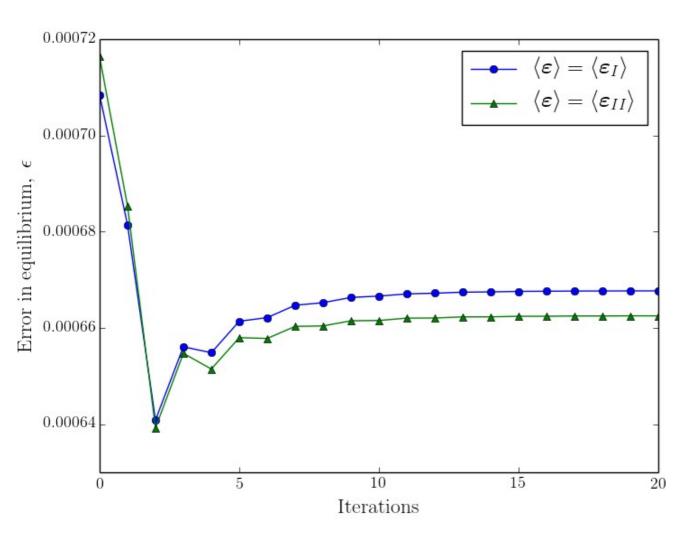
Error in equilibrium:

$${}^{m}\epsilon = \left[\frac{\langle \|^{m}\widehat{\boldsymbol{\sigma}}(\underline{k})\,\underline{k}\|^{2}\rangle}{{}^{m}\widehat{\boldsymbol{\sigma}}(\underline{0}):{}^{m}\widehat{\boldsymbol{\sigma}}(\underline{0})}\right]^{1/2}$$

- (1) Convergence depends on contrast between phases, and \mathbb{C}_0 .
- (2) Can not solve for infinite contrasts, i.e. rigid/compliant phases.
- (3) Numerical homogenization requires several numerical tests.

Convergence – Issues encountered

1) Increasing error vs number of iterations (also observed by Lebensohn, 2001).



$$N_x \times N_y = 2000 \times 2000$$

$$\mathbb{C}_0 = \langle \mathbb{C} \rangle$$

$$\langle \boldsymbol{\varepsilon}_I \rangle = 10^{-5} \underline{e}_1 \otimes \underline{e}_1$$

$$\langle \boldsymbol{\varepsilon}_{II} \rangle = 10^{-5} \underline{e}_2 \otimes \underline{e}_2$$

2) Some cases of instability, e.g. isotropic matrix-fiber w/ much stiffer fiber.

Numerical homogenization (strain-based)

We solve for $\widetilde{\mathbb{C}}$ such that $\langle \boldsymbol{\sigma} \rangle = \widetilde{\mathbb{C}} : \langle \boldsymbol{\varepsilon} \rangle$. From the Hill-Mandel principle, we have:

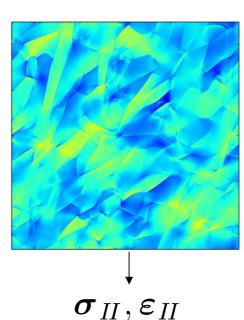
$$\langle \boldsymbol{\sigma} : \boldsymbol{\varepsilon} \rangle = \widetilde{C}_{1111} \langle \varepsilon_{11} \rangle^2 + \widetilde{C}_{2222} \langle \varepsilon_{22} \rangle^2 + 4\widetilde{C}_{1212} \langle \varepsilon_{12} \rangle^2$$

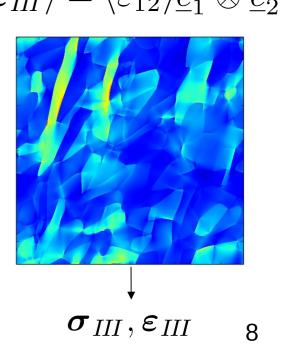
$$+ 2\widetilde{C}_{1122} \langle \varepsilon_{11} \rangle \langle \varepsilon_{22} \rangle + 4\widetilde{C}_{1112} \langle \varepsilon_{11} \rangle \langle \varepsilon_{12} \rangle$$

$$+ 4\widetilde{C}_{2212} \langle \varepsilon_{22} \rangle \langle \varepsilon_{12} \rangle$$

Consider three mean-field loading cases:

$$\langle arepsilon_I \rangle = \langle arepsilon_{11}
angle \underline{e}_1 \otimes \underline{e}_1 \quad \langle arepsilon_{II}
angle = \langle arepsilon_{22}
angle \underline{e}_2 \otimes \underline{e}_2 \quad \langle arepsilon_{III}
angle = \langle arepsilon_{12}
angle \underline{e}_1 \otimes \underline{e}_2$$





Numerical homogenization (strain-based)

Then, from these three loading cases, we have:

$$\widetilde{C}_{1111} = \frac{\langle \boldsymbol{\sigma}_I : \boldsymbol{\varepsilon}_I \rangle}{\langle \varepsilon_{11} \rangle^2} \qquad \widetilde{C}_{2222} = \frac{\langle \boldsymbol{\sigma}_{II} : \boldsymbol{\varepsilon}_{II} \rangle}{\langle \varepsilon_{22} \rangle^2} \qquad \widetilde{C}_{1212} = \frac{\langle \boldsymbol{\sigma}_{III} : \boldsymbol{\varepsilon}_{III} \rangle}{4\langle \varepsilon_{12} \rangle^2}$$

Also, by superposition, we have:

$$\widetilde{C}_{1122} = \frac{\langle (\boldsymbol{\sigma}_{I} + \boldsymbol{\sigma}_{II}) : (\boldsymbol{\varepsilon}_{I} + \boldsymbol{\varepsilon}_{II}) \rangle - \widetilde{C}_{1111} \langle \boldsymbol{\varepsilon}_{11} \rangle^{2} - \widetilde{C}_{2222} \langle \boldsymbol{\varepsilon}_{22} \rangle^{2}}{2 \langle \boldsymbol{\varepsilon}_{11} \rangle \langle \boldsymbol{\varepsilon}_{22} \rangle}$$

$$\widetilde{C}_{1112} = \frac{\langle (\boldsymbol{\sigma}_I + \boldsymbol{\sigma}_{III}) : (\boldsymbol{\varepsilon}_I + \boldsymbol{\varepsilon}_{III}) \rangle - \widetilde{C}_{1111} \langle \boldsymbol{\varepsilon}_{11} \rangle^2 - 4\widetilde{C}_{1212} \langle \boldsymbol{\varepsilon}_{12} \rangle^2}{4 \langle \boldsymbol{\varepsilon}_{11} \rangle \langle \boldsymbol{\varepsilon}_{12} \rangle}$$

$$\widetilde{C}_{2212} = \frac{\langle (\boldsymbol{\sigma}_{II} + \boldsymbol{\sigma}_{III}) : (\boldsymbol{\varepsilon}_{II} + \boldsymbol{\varepsilon}_{III}) \rangle - \widetilde{C}_{2222} \langle \varepsilon_{22} \rangle^2 - 4\widetilde{C}_{1212} \langle \varepsilon_{12} \rangle^2}{4 \langle \varepsilon_{22} \rangle \langle \varepsilon_{12} \rangle}$$

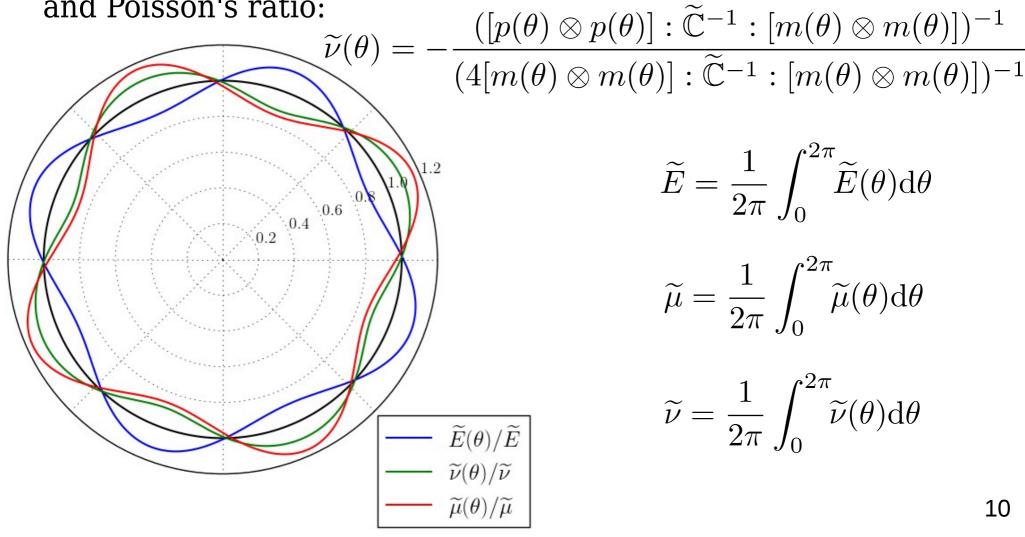
Moduli for elastic anisotropy

Following Hayes (1972), we have the following moduli:

$$\widetilde{E}(\theta) = ([m(\theta) \otimes m(\theta)] : \widetilde{\mathbb{C}}^{-1} : [m(\theta) \otimes m(\theta)])^{-1}$$

$$\widetilde{\mu}(\theta) = (4[m(\theta) \otimes p(\theta)] : \widetilde{\mathbb{C}}^{-1} : [m(\theta) \otimes p(\theta)])^{-1}$$

and Poisson's ratio:



$$\widetilde{E} = \frac{1}{2\pi} \int_{0}^{2\pi} \widetilde{E}(\theta) d\theta$$

$$\widetilde{\mu} = \frac{1}{2\pi} \int_0^{2\pi} \widetilde{\mu}(\theta) d\theta$$

$$\widetilde{\nu} = \frac{1}{2\pi} \int_{0}^{2\pi} \widetilde{\nu}(\theta) d\theta$$

Mean-field elastic self-consistent homogenization

The following interaction law is assumed between the overall mean strain state $\langle \varepsilon \rangle$ and each local mean field $\langle \varepsilon \rangle_{\alpha}$:

$$\langle arepsilon
angle_lpha = \widehat{\mathbb{T}}_lpha : \langle arepsilon
angle$$

where $\widehat{\mathbb{T}}_{\alpha}$ is a strain concentration tensor given by

$$\widehat{\mathbb{T}}_{\alpha} = [\mathbb{I} + \widehat{\mathbb{P}}^{\alpha} : (\mathbb{C}_{\alpha} - \widehat{\mathbb{C}})]^{-1}$$

The Hill-polarization tensor $\widehat{\mathbb{P}}_{\alpha}$ is computed after the assumption that Ω_{α} is idealized as an ellipsoid $\widehat{\Omega}_{\alpha}$:

$$\widehat{\mathbb{P}^{\alpha}} = \widehat{\mathbb{P}^{\alpha}}(\underline{x}) = \int_{\widehat{\Omega}_{\alpha}} \mathcal{F}^{-1} \left\{ \underline{\xi} \odot [\underline{\xi} \, \widehat{\mathbb{C}} \, \underline{\xi}]^{-1} \odot \underline{\xi} \right\} (\underline{x} - \underline{y}) \mathrm{d}V_{\underline{y}}$$
Computed after Masson (2008)

The apparent stiffness is solved for iteratively from:

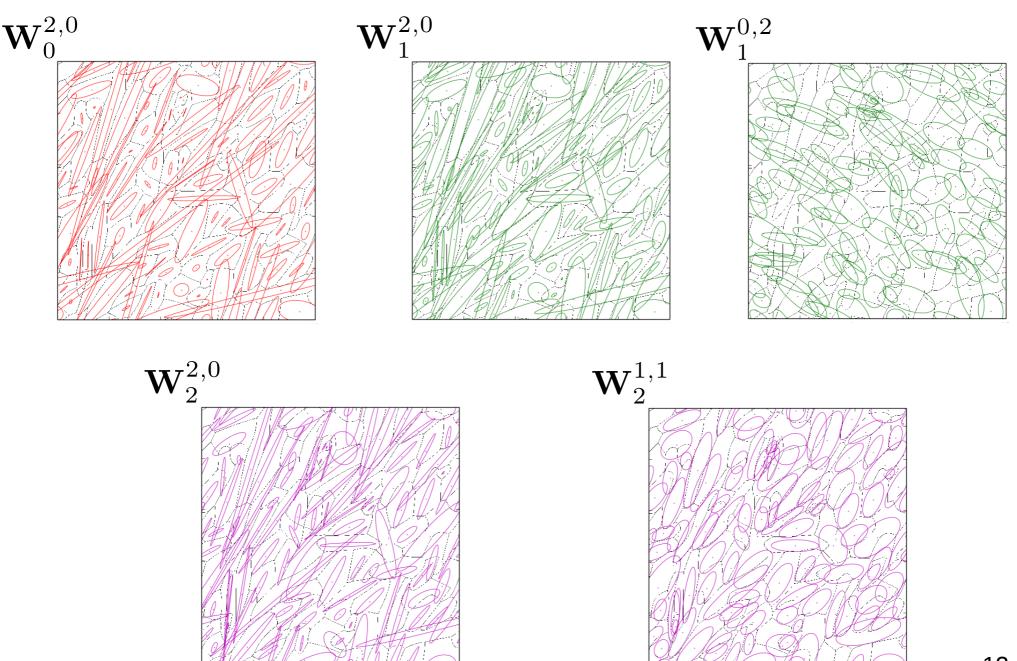
$$\widehat{\mathbb{C}} = \sum_{\alpha=1}^{n} c_{\alpha} \mathbb{C}_{\alpha} : \widehat{\mathbb{T}}_{\alpha}$$

An arbitrary Ω_{α} can be idealized by many different ellipsoids $\widehat{\Omega}_{\alpha}$.

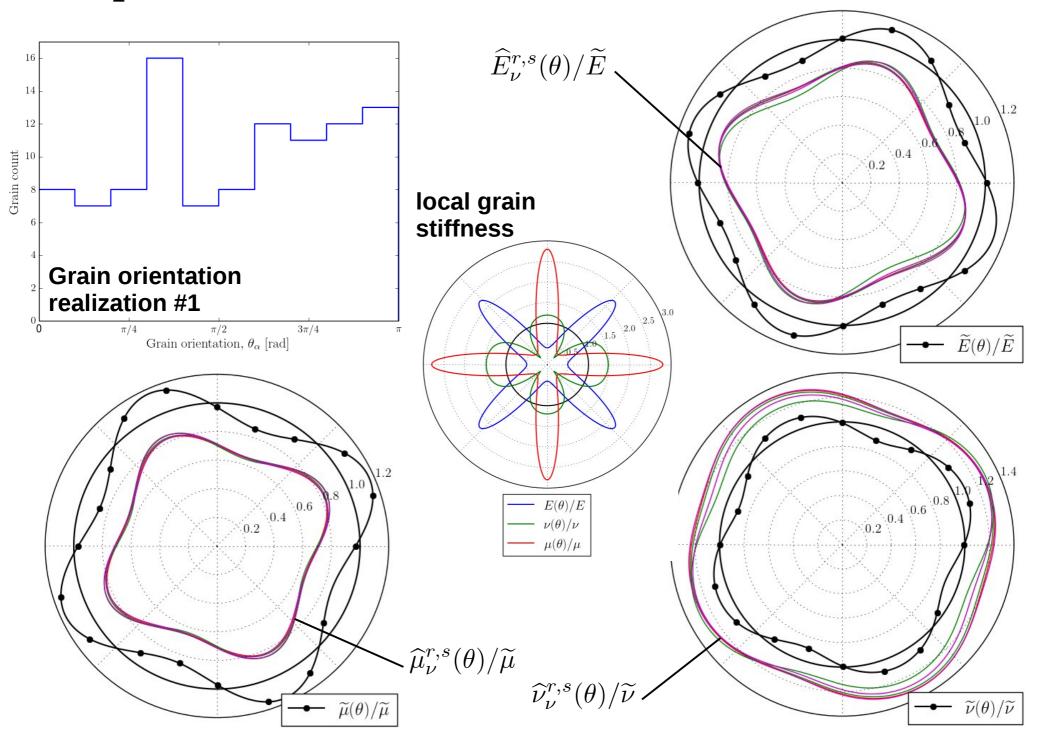
What morphological idealization gives best estimates $\widehat{\mathbb{C}}$?

Different morphological idealizations

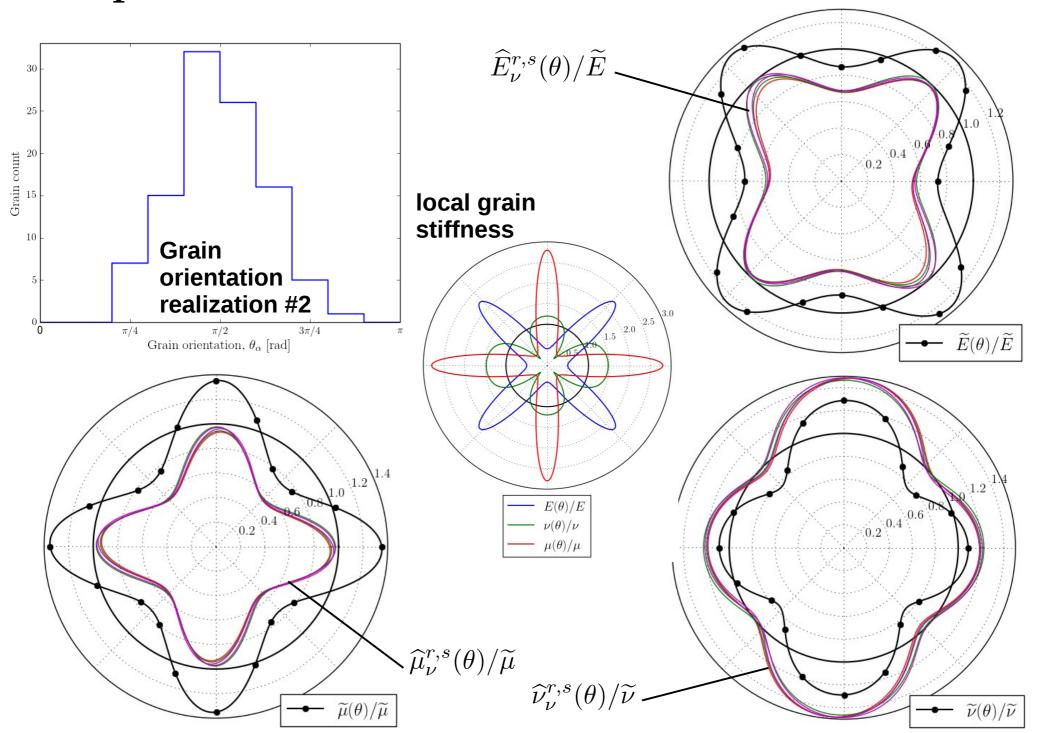
From each characterization with 2nd order Minkowski tensors:



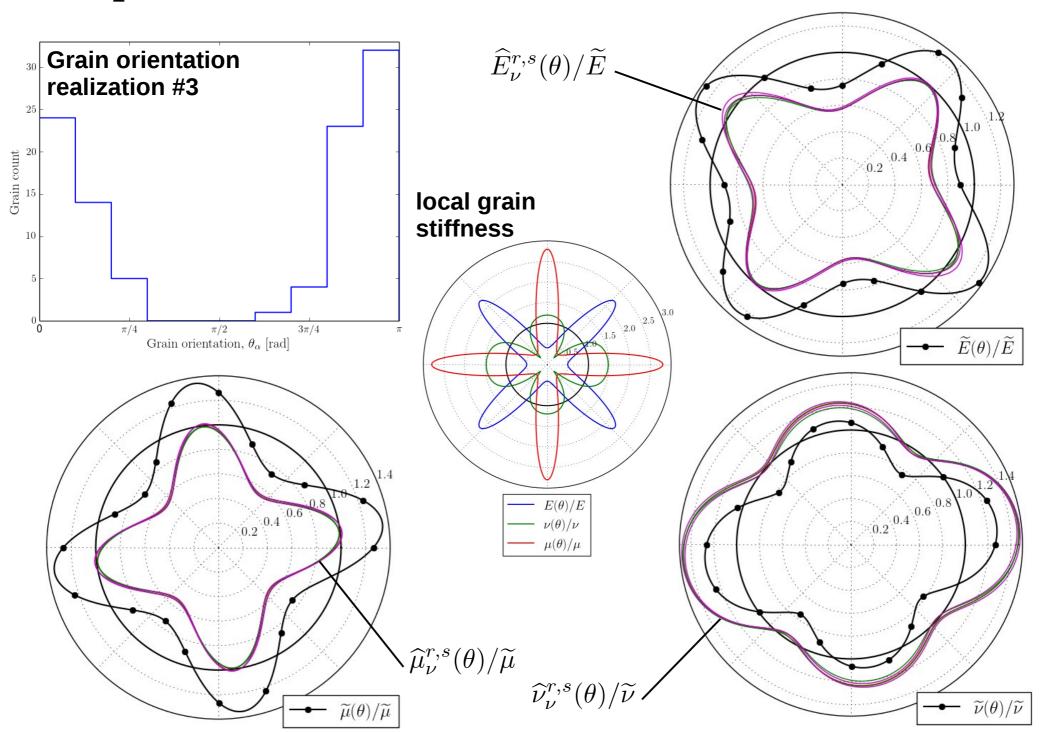
Comparison of SC-based elastic stiffness estimates



Comparison of SC-based elastic stiffness estimates



Comparison of SC-based elastic stiffness estimates



Conclusion

- Morphological characterization of single grains
- Numerical homogenization
- Self-consistent homogenization
- Shape idealization
- Comparison of SC estimates of elastic stiffness

References

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