# Computational mechanics and stochastic simulation of random polycrystals

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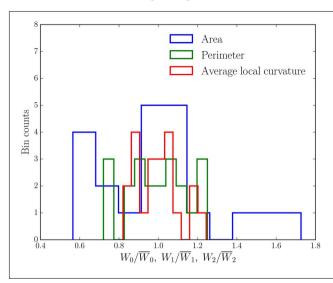


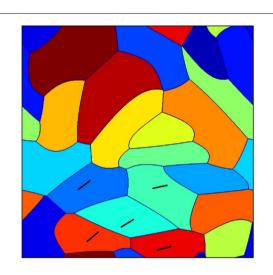
Graduate Board Oral Exam December 11, 2017

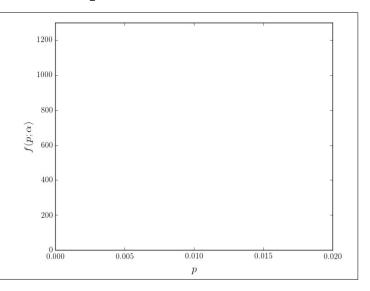
#### Different morphologies lead to different mechanical behaviors.

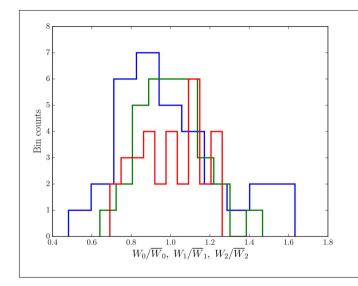
Morphological statistics of single grains

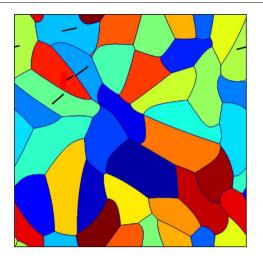
Periodic viscoplastic polycrystal subjected to  $\overline{\varepsilon}(t) \propto t\underline{e}_1 \otimes \underline{e}_1$ 

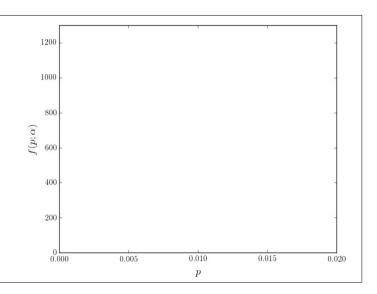








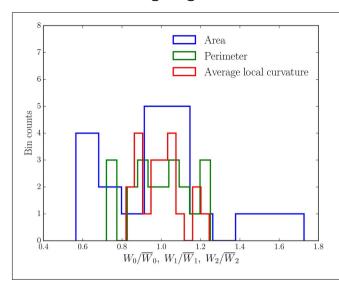


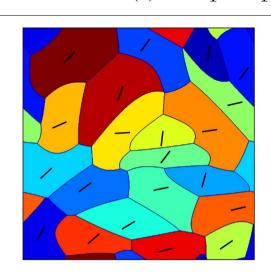


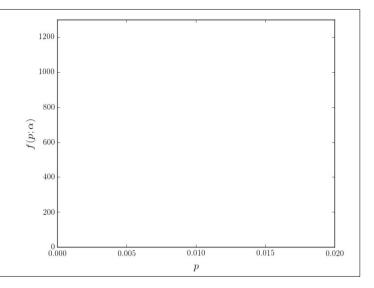
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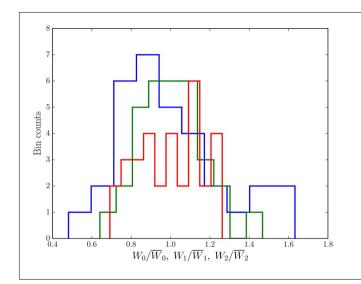
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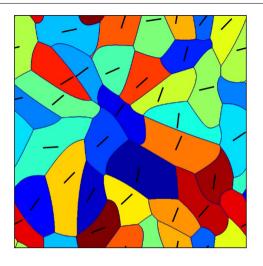
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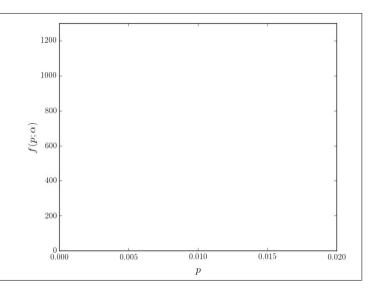








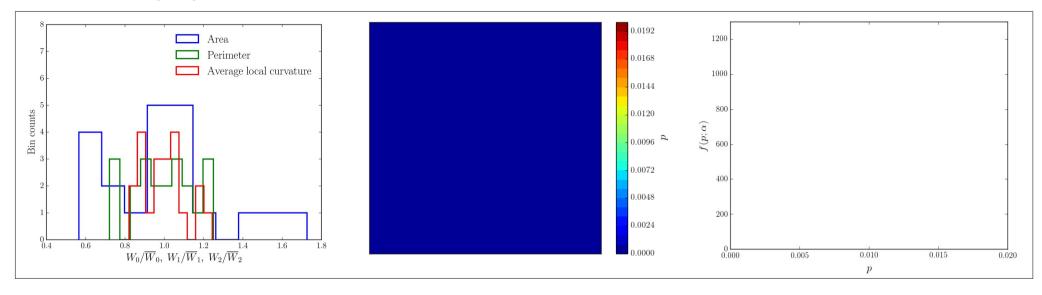


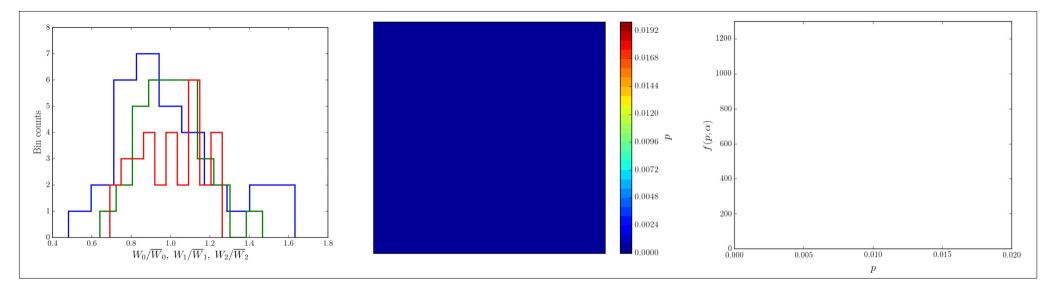


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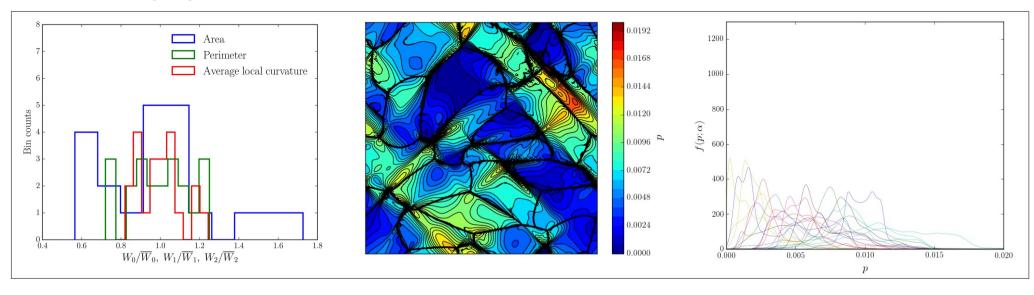


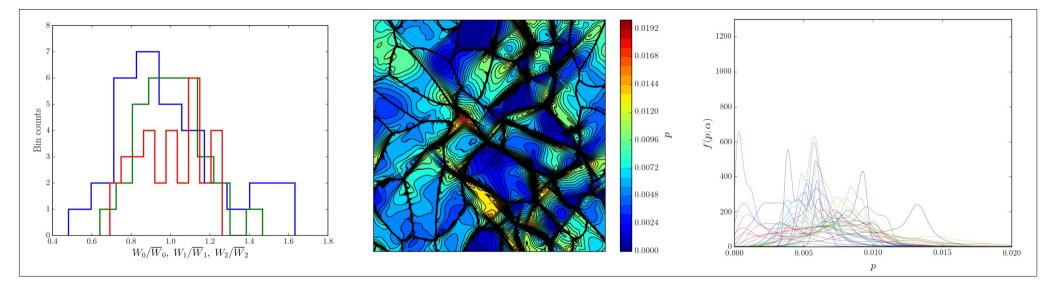


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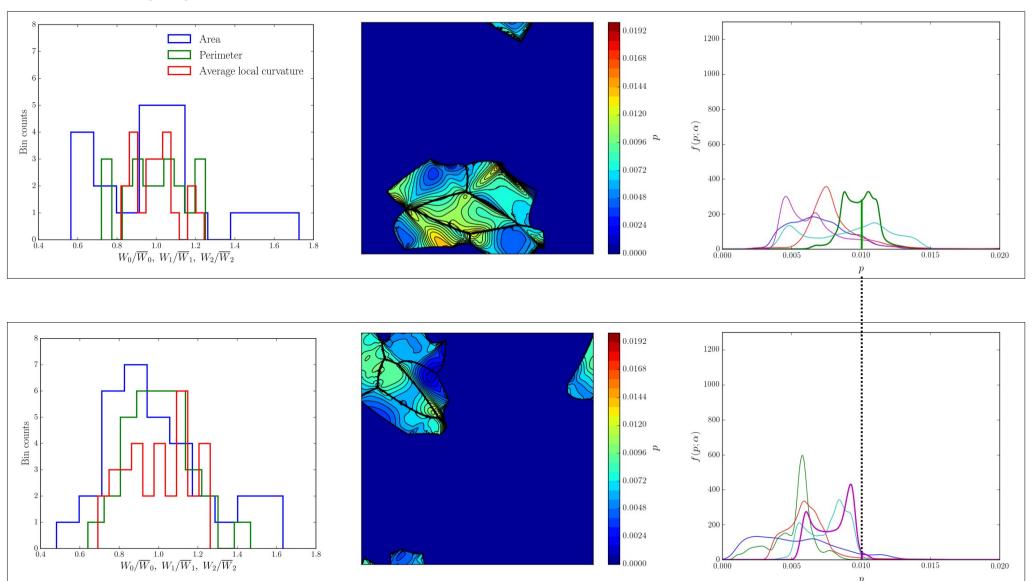




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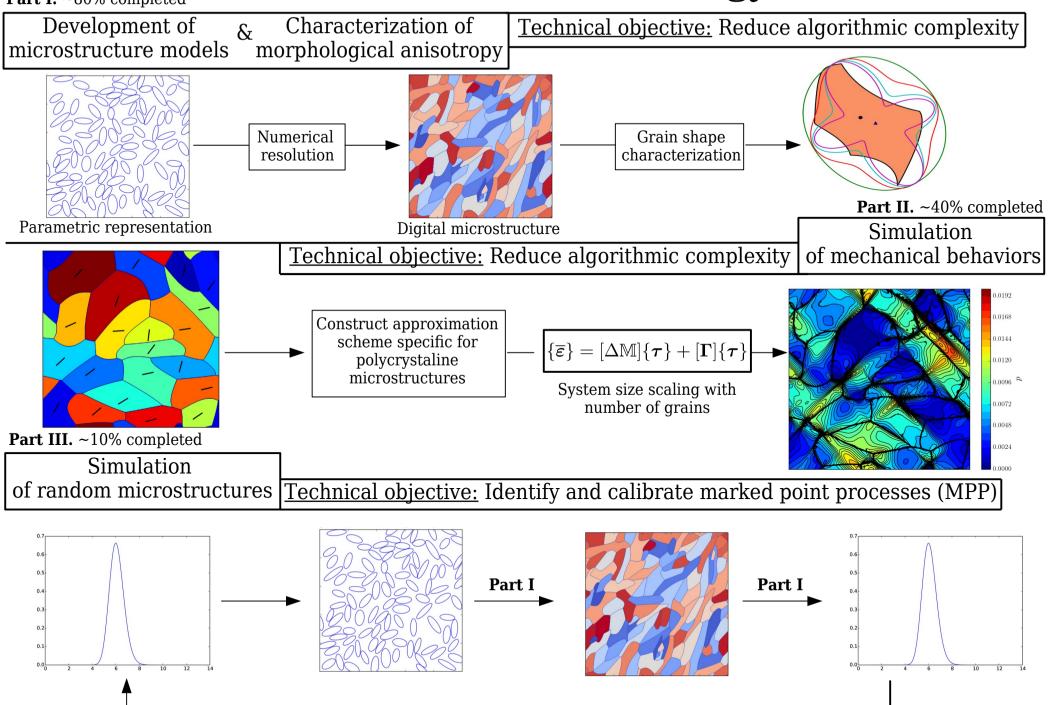
Morphological statistics of single grains

Periodic viscoplastic polycrystal subjected to  $\overline{\varepsilon}(t) \propto t\underline{e}_1 \otimes \underline{e}_1$ 



#### Part I. ~80% completed

# Outline of methodology



# Part I. Parametric representation

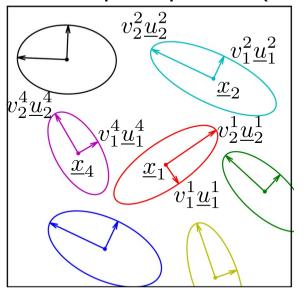
Consider a pattern of  $n_{\alpha}$  marked points

$$\Xi = \{(\underline{x}_{\alpha}, \mathbf{Z}_{\alpha}) \mid 1 \le \alpha \le n_{\alpha}, \ \mathbf{Z}_{\alpha} \succ 0\}$$

The underlying microstructure  $T_{\rm ess}(\Xi)$  is a partition of space into  $n_{\alpha}$  cells (or grains)

$$\Omega_{\alpha} = \{ \underline{x} | \underset{\gamma}{\operatorname{argmin}} (\underline{x} - \underline{x}_{\gamma}) \cdot \mathbf{Z}_{\gamma} \cdot (\underline{x} - \underline{x}_{\gamma}) = \alpha \}$$

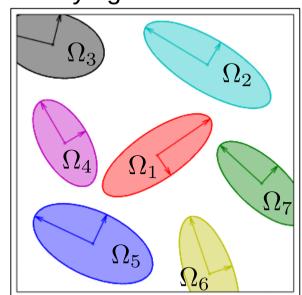
Marked point pattern (MPP)



$$\mathbf{Z}_{\alpha} = \frac{\underline{u}_{1}^{\alpha} \otimes \underline{u}_{1}^{\alpha}}{(v_{1}^{\alpha})^{2}} + \frac{\underline{u}_{2}^{\alpha} \otimes \underline{u}_{2}^{\alpha}}{(v_{2}^{\alpha})^{2}}$$

Numerical Resolution

#### Underlying microstructure



# Part I. Parametric representation

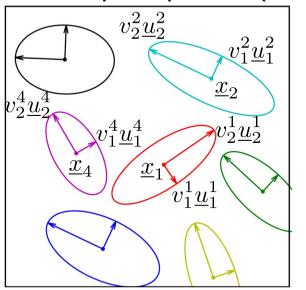
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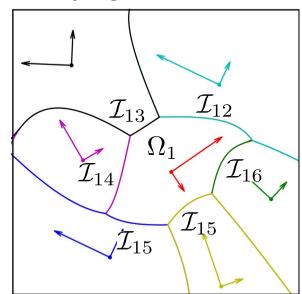
Marked point pattern (MPP)



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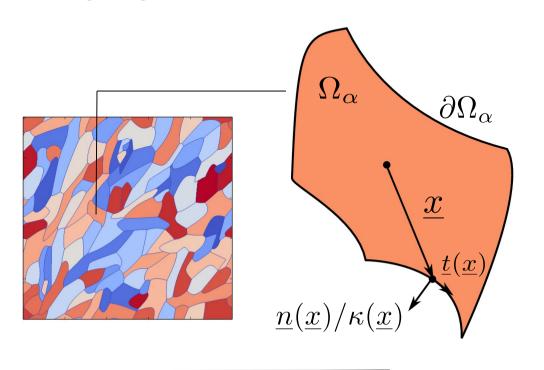
Underlying microstructure



Every cell  $\Omega_{\alpha}$  has a boundary  $\partial\Omega_{\alpha}$  partitioned into common curves  $\mathcal{I}_{\alpha\gamma}$  shared with neighbors its  $\Omega_{\gamma}$ .

# Part I. Morphological characterization

Single grains are characterized using Minkowski tensors:



Measures of mass distribution:

$$\mathcal{W}_0^{r,0} = \int_{\Omega_\alpha} \underline{x}^{\otimes^r} \mathrm{d}V$$

Measures of surface distribution:

$$\mathcal{W}_1^{r,s} = \int_{\partial\Omega_x} \underline{x}^{\otimes^r} \odot [\underline{n}(\underline{x})]^{\otimes^s} dS$$

<u>Curvature-weighted measures of</u> surface distribution:

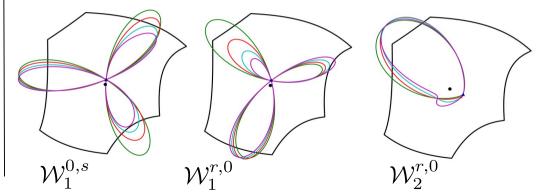
$$\mathcal{W}_{2}^{r,s} = \int_{\partial\Omega_{\alpha}} \kappa(\underline{x}) \underline{x}^{\otimes^{r}} \odot [\underline{n}(\underline{x})]^{\otimes^{s}} dS$$

Examples of projection up to order 8

$$\begin{array}{c} - : r + s = 2 \\ - : r + s = 4 \\ \hline \\ \vdots \\ r + s = 8 \\ \hline \end{array}$$

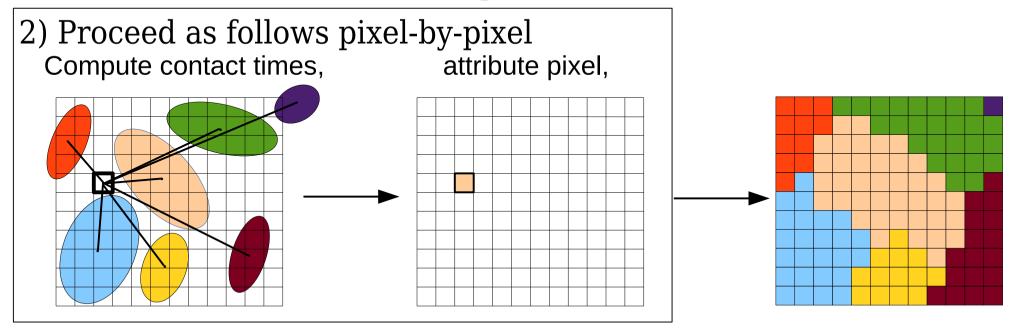
$$---: r+s=3$$
  $---: r+s=7$ 

$$---: r+s=5$$
  $---: r+s=9$ 

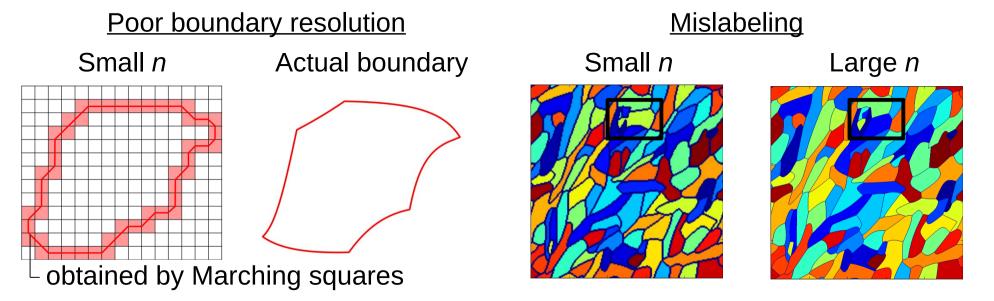


# Part I. Numerical resolution – naive approach

1) Discretize the domain into *n* pixels

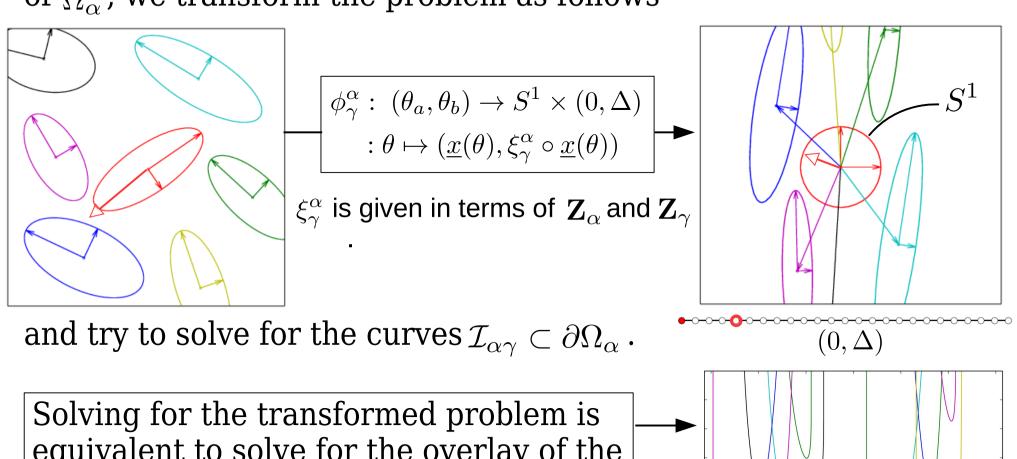


Resulting algorithm has complexity O(n). How large need n be?



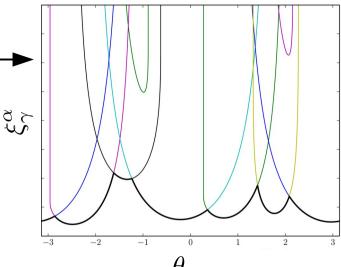
# Part I. Transformation of the problem

Instead of discretizing the plane and solving for approximations of  $\Omega_{\alpha}$ , we transform the problem as follows



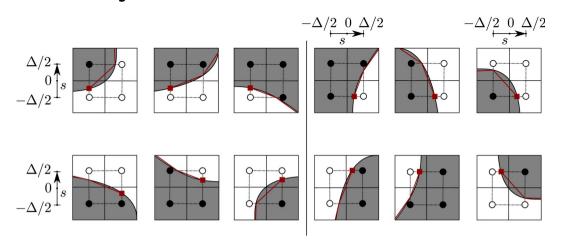
equivalent to solve for the overlay of the functions  $\xi^{\alpha}_{\gamma}$  of the neighbors  $\Omega_{\gamma}$  of  $\Omega_{\alpha}$ .

But, what are the neighbors of  $\Omega_{\alpha}$ ?



# Part I. Enriched Marching Squares (EMS)

Using our analytical solution for the grain boundary, we obtain an enriched MS (EMS)



# MS

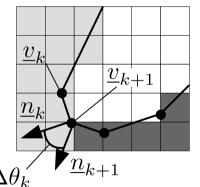
#### Expressions of Minkowski tensors:

$$\mathcal{W}_{0}^{r,0} = \frac{1}{r+2} \sum_{k=1}^{n} \sum_{i=0}^{r+1} \sum_{j=0}^{i} {r+1 \choose i} {i \choose j} \frac{(-1)^{i-j} L_{k}}{i+1} \left[ \underline{v}_{k}^{\otimes^{r+1-j}} \odot \underline{v}_{k+1}^{\otimes^{j}} \right] \cdot \underline{n}_{k},$$

$$\mathcal{W}_{1}^{r,s} = \frac{1}{2} \sum_{k=1}^{n} \sum_{i=0}^{r} \sum_{j=0}^{i} {r \choose i} {i \choose j} \frac{(-1)^{i-j} L_{k}}{i+1} \underline{v}_{k}^{\otimes^{r-j}} \odot \underline{v}_{k+1}^{\otimes^{j}} \odot \underline{n}_{k}^{\otimes^{s}},$$

$$\mathcal{W}_{2}^{r,s} = \frac{1}{2} \sum_{k=1}^{n} \sum_{i=0}^{s} \sum_{j=0}^{i} {s \choose i} {i \choose j} \frac{(-1)^{i-j} L_{k}}{L_{k}^{i}} \underline{v}_{k}^{\otimes^{i-j}} \odot \underline{v}_{k+1}^{\otimes^{r+j}} \odot \underline{n}_{k}^{\otimes^{s-i}} I_{k}^{s,i} (\Delta \theta_{k})$$

$$L_k = \|\underline{v}_{k+1} - \underline{v}_k\|$$



# Part I. Expressions of Minkowski tensors

If the expressions of the common curves and the common points

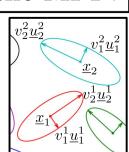
are known, there is no need to approximate the boundaries using an MS algorithm. We have

 $\underline{x}_{\alpha}, \ \underline{u}_{1}^{\alpha} \ \text{and} \ \underline{u}_{2}^{\alpha}$ are from the MPP.

common points

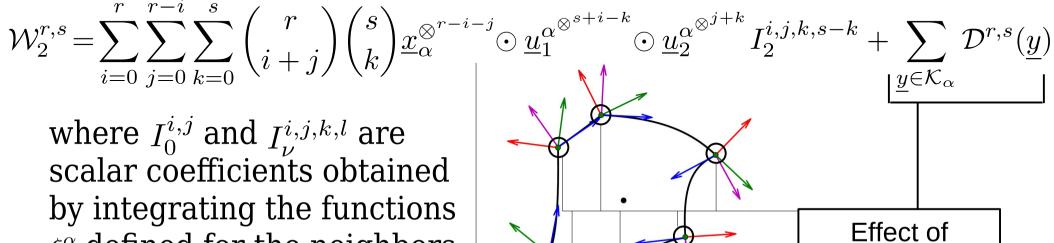
$$\mathcal{W}_{0}^{r,0} = \sum_{i=0}^{r} \sum_{j=0}^{r-i} \binom{r}{i+j} \underline{x}_{\alpha}^{\otimes^{r-i-j}} \odot \underline{u}_{1}^{\alpha^{\otimes^{i}}} \odot \underline{u}_{2}^{\alpha^{\otimes^{j}}} I_{0}^{i,j}$$

$$\mathcal{W}_{1}^{r,s} = \sum_{i=0}^{r} \sum_{j=0}^{r-i} \sum_{k=0}^{s} {r \choose i+j} {s \choose k} \underline{x}_{\alpha}^{\otimes^{r-i-j}} \odot \underline{u}_{1}^{\alpha^{\otimes^{s+i-k}}} \odot \underline{u}_{2}^{\alpha^{\otimes^{j+k}}} I_{1}^{i,j,k,s-k}$$



$$\mathcal{W}_{2}^{r,s} = \sum_{i=0}^{r} \sum_{j=0}^{r-i} \sum_{k=0}^{s} \binom{r}{i+j} \binom{s}{k} \underline{x}_{\alpha}^{\otimes^{r-i}}$$

where  $I_0^{i,j}$  and  $I_n^{i,j,k,l}$  are scalar coefficients obtained by integrating the functions  $\xi^{\alpha}_{\gamma}$  defined for the neighbors  $\Omega_{\gamma}$  of  $\Omega_{\alpha}$ .



# Part II. Periodic Lippmann-Schwinger problems

Periodic elastic Cauchy-Navier problem:

$$\begin{split} & \boldsymbol{\sigma}(\underline{x}) = \mathbb{L}(\underline{x}) : \boldsymbol{\varepsilon}(\underline{x}) \;, & \nabla \cdot \boldsymbol{\sigma}(\underline{x}) = \underline{0} \;, & \boldsymbol{\varepsilon}(\underline{x}) = \{\nabla \underline{u}(\underline{x})\}_{sym} \\ & \text{with } \mathbb{L}(\underline{x} + (n\underline{e}_1 + m\underline{e}_2)L) = \mathbb{L}(\underline{x}) \text{ for all } \underline{x} \in \mathbb{R}^2 \text{ and } n, m \in \mathbb{Z} \;, \\ & \text{and } & \boldsymbol{\sigma}(\underline{x} + (n\underline{e}_1 + m\underline{e}_2)) \cdot \underline{e}_k = \boldsymbol{\sigma}(\underline{x}) \cdot \underline{e}_k \;, \\ & \text{subjected to } \overline{\boldsymbol{\varepsilon}} = \boldsymbol{\varepsilon}_0. \end{split}$$

where 
$$lacksquare = rac{1}{L^2} \!\! \int_{\Omega} \!\! ullet(\underline{x}) \mathrm{d} 
u_{\underline{x}} \ \, \mathrm{and} \ \, \Omega \! := \! [0,L] imes [0,L].$$

Introduce a polarization field au with reference  $\mathbb{L}^0$ ,

$$\boldsymbol{\tau}(\underline{x}) := \boldsymbol{\sigma}(\underline{x}) - \mathbb{L}^0 : \boldsymbol{\varepsilon}(\underline{x}) = [\mathbb{L}(\underline{x}) - \mathbb{L}^0] : \boldsymbol{\varepsilon}(\underline{x}) = \Delta \mathbb{L}(\underline{x}) : \boldsymbol{\varepsilon}(\underline{x})$$

leads to a <u>Periodic elastic Lippmann-Schwinger problem</u>:

$$abla \cdot oldsymbol{ au}(\underline{x}) + 
abla \cdot [\mathbb{L}^0 : oldsymbol{arepsilon}(\underline{x})] = \underline{0}$$
 Auxiliary pb.

with solution

$$\boldsymbol{\varepsilon}(\underline{x}) = \overline{\boldsymbol{\varepsilon}} - \mathbf{\Gamma} * \boldsymbol{\tau}(\underline{x}) = \overline{\boldsymbol{\varepsilon}} - \mathbf{\Gamma} * [\Delta \mathbb{L} : \boldsymbol{\varepsilon}(\underline{x})]$$

Periodic Green operator for strains.

where  $\mathbf{\Gamma} * \boldsymbol{\tau}(\underline{x}) := \int_{\mathbb{R}^2} \mathbf{\Gamma}(\underline{x}' - \underline{x}) : \boldsymbol{\tau}(\underline{x}') \; \mathrm{d}\nu_{\underline{x}'}$  is a convolution of  $\mathbf{\Gamma}(\underline{\Delta}\underline{x})$ .

# Part II. Basic approximation scheme

FFT-based iterative scheme:

$$\begin{aligned} & \boldsymbol{\tau}(\underline{x}) \leftarrow \Delta \mathbb{L}(\underline{x}) : \boldsymbol{\varepsilon}_{0} \\ & {}^{0}\boldsymbol{\tau}(\underline{x}) \leftarrow \boldsymbol{\tau}(\underline{x}) \\ & \text{while } \| \overline{\nabla \cdot \boldsymbol{\sigma}} \| > tol : \\ & \hat{\tilde{\boldsymbol{\varepsilon}}}(\underline{0}) \leftarrow \mathbf{0} \\ & \hat{\tilde{\boldsymbol{\varepsilon}}}(\underline{\omega}) \leftarrow -\hat{\boldsymbol{\Gamma}}(\underline{\omega}) : \mathcal{F}\mathcal{F}\mathcal{T}\{\boldsymbol{\tau}(\underline{x})\}(\underline{\omega}) \ \forall \ \underline{\omega} \neq \underline{0} \\ & \boldsymbol{\tau}(\underline{x}) \leftarrow {}^{0}\boldsymbol{\tau}(\underline{x}) + \Delta \mathbb{L}(\underline{x}) : \mathcal{F}\mathcal{F}\mathcal{T}^{-1}\{\hat{\tilde{\boldsymbol{\varepsilon}}}(\underline{\omega})\}(\underline{x}) \end{aligned}$$

#### Instead, we want a scheme that

- is especially tailored to polycrystalline microstructures,
- has complexity governed by the number of grains
- carries information about the morphology
- is easy to post-process.

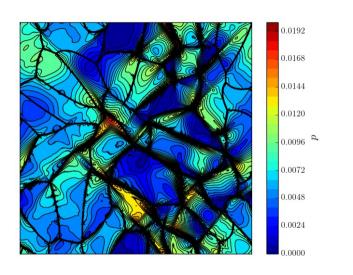
#### Pros:

- Easy pre-processing,
- Compatible with an elastic eigenstrain formulations.

#### Cons:

- Uses a lot of memory,
- Post-processing is data intensive,
- No explicit insight about morphology,
- Complexity is  $O(n \log n)$  per iteration, for n pixels.

How large need *n* be? How many iterations are enough?



# Part II. Hashin-Shtrikman variational principle

Consider the Hashin-Shtrikman (HS) functional given by

$$2\mathcal{H}(\boldsymbol{\tau}) := \boldsymbol{\varepsilon}_0 : \mathbb{L}_0 : \boldsymbol{\varepsilon}_0 + \overline{(2\boldsymbol{\tau} : \boldsymbol{\varepsilon}_0 - \boldsymbol{\tau} : (\Delta \mathbb{L})^{-1} : \boldsymbol{\tau} - \boldsymbol{\tau} : (\boldsymbol{\Gamma} * \boldsymbol{\tau}))}$$

- (1) Assuming an equilibrated polarization, i.e.  $\varepsilon(\underline{x}) = \overline{\varepsilon} \Gamma * \tau(\underline{x})$ ,  $\mathcal{H}(\tau)$  is stationary at  $\tau(\underline{x}) = \Delta \mathbb{L} : \varepsilon(\underline{x})$ , irrespectively of  $\mathbb{L}_0$ :

  The solution to the Lippmann-Schwinger problem is a stationary point of the HS functional.
- (2) Hashin and Shtrikman (1962) proved that

$$\Delta \mathbb{L}(\underline{x}) \prec 0 \implies \mathcal{H}(\boldsymbol{\tau})$$
 is strictly convex

Okay. Then, let's consider a piecewise polynomial ansatz

$$m{ au}^{h_p}(\underline{x}) := \sum_{lpha} \chi_{lpha}(\underline{x}) \sum_{k=0}^p \left\langle m{ au}^{lpha} m{\partial}^k, (\underline{x} - \underline{x}^{lpha})^{\otimes^k} 
ight
angle_k$$

Symmetric tensor of order k+2

for some p and solve for the minimizer  $\{\tau^{\alpha}\partial^{k}, 0 \leq k \leq p\}$  of  $\mathcal{H}(\tau^{h_p})$ .

# Part II. Implementation & validation

Note that  $\mathcal{H}(\boldsymbol{\tau}^{h_p})$  is a quadratic form. Consequently, the minimizer  $\{\boldsymbol{\tau}^{\alpha}\boldsymbol{\partial}^k, 0 \leq k \leq p\}$  is solution of the linear system

$$\{\overline{oldsymbol{arepsilon}}\} = [\Delta \mathbb{M}]\{oldsymbol{ au}\} + [oldsymbol{\Gamma}]\{oldsymbol{ au}\}$$

$$N \times 1$$
  $N \times N$   $N \times 1$   $N \times N$   $N \times 1$  where  $N = 3n_{\alpha} \binom{p+2}{2}$ .

in which  $[\Delta \mathbb{M}]_{ij} \propto \text{components of } \Delta \mathbb{M}^{\alpha} \otimes \mathcal{W}_0^{r,0}(\Omega_{\alpha})$  with  $r \leq 2p$ 

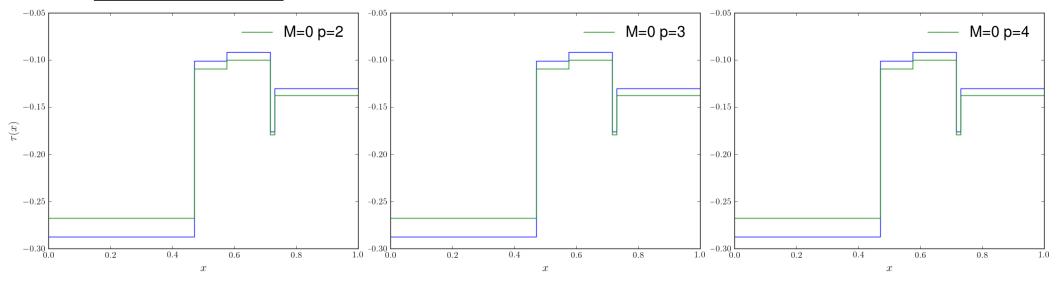
with  $r \leq 2p$ 

(part I)

 $[\Gamma]_{ij} \propto \text{components of weighted integrals of } \hat{\Gamma}(\underline{\omega})$  evaluated at 2M wave numbers

So that the same  $[\Gamma]$  can be used for several realizations of  $\Delta \mathbb{L}(\underline{x})$ .

#### 1D validation:



# References

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# Extra-Slide #2 – Transform the problem

Solving for parameterizations of common curves  $\mathcal{I}_{\alpha\gamma}$  is difficult. To circumvent this difficulty, we introduce a diffeomorphic transformation.

Let every point of a growing ellipse be given by a time-

dependent mapping from a unit circle:

$$\varphi_{\alpha}: S^1 \times (0, \Delta) \to S_{\alpha} \subset \mathbb{R}^2$$

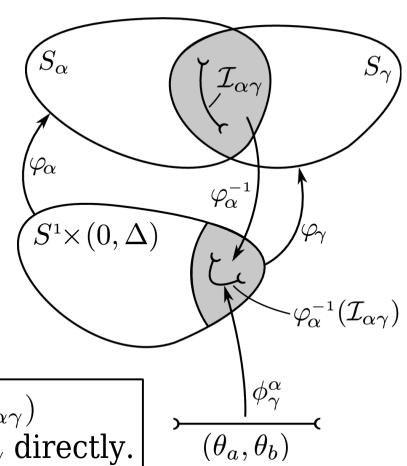
$$: (\underline{x}, t) \mapsto \underline{x}_{\alpha} + t \mathbf{Z}_{\alpha}^{-1/2} \cdot \underline{x}$$

We let the common curves be

$$\mathcal{I}_{\alpha\beta} = \{ y \in S_{\alpha} \cap S_{\gamma} \mid f_{\gamma}^{\alpha}(y) = 0 \}$$

with 
$$f_{\gamma}^{\alpha}(\underline{y}) = \tau \circ \varphi_{\alpha}^{-1}(\underline{y}) - \tau \circ \varphi_{\gamma}^{-1}(\underline{y})$$
.

Finding parameterizations  $\phi_{\gamma}^{\alpha}$  of  $\varphi_{\alpha}^{-1}(\mathcal{I}_{\alpha\gamma})$  is much easier than parameterizing  $\mathcal{I}_{\alpha\gamma}$  directly.



# Extra-slide #1 – Viscoplastic polycrystal model

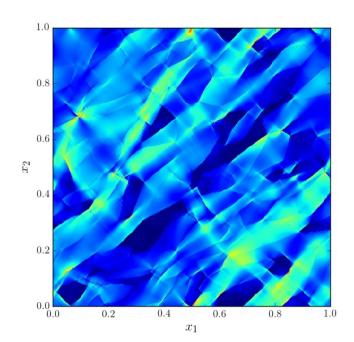
• Constitutive model:  $\dot{\boldsymbol{\sigma}}(t) = \mathbb{L} : [\dot{\boldsymbol{\varepsilon}}(t) - \dot{\boldsymbol{\varepsilon}}^p(t)]$ 

$$\dot{\boldsymbol{\varepsilon}}^{p} = \sum_{\alpha=1}^{2} \dot{\gamma}^{(\alpha)} \boldsymbol{\mu}^{(\alpha)} \qquad \boldsymbol{\mu}^{(\alpha)} = \underline{m}_{\alpha} \overset{s}{\otimes} \underline{n}_{\alpha}$$

$$\dot{\gamma}^{(\alpha)} = \dot{\gamma}_{0}^{(\alpha)} \left( \frac{|\boldsymbol{\sigma} : \boldsymbol{\mu}^{(\alpha)}|}{\tau^{\alpha}} \right)^{\frac{1}{m}} \operatorname{sgn}(\boldsymbol{\sigma} : \boldsymbol{\mu}^{(\alpha)}) \quad \dot{\tau}^{\alpha} = h \sum_{\beta=1}^{2} |\dot{\gamma}^{(\alpha)}|$$

- Material properties: 2D isotropic stiffness;
- Sources of randomness:
  - Grain morphology,
     lattice misorientation

• Quantity of interest:  $p = \sum_{\alpha=1}^{z} \int_{0}^{t} |\dot{\gamma}^{(\alpha)}| d\tau$ 



Part III. Realization dependent conditional intensity

Markov Marked point processes are an ideal choice of model.